



## Defect formation and the water–gas shift reaction on $\beta$ -Ga<sub>2</sub>O<sub>3</sub>

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### ABSTRACT

The water–gas shift reaction was studied in either direction on  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> samples with different surface chemistries: a fully oxidized surface, a hydrogen-pretreated surface with and without oxygen vacancies present, and an oxygen-defective surface without adsorbed hydrogen. The samples were characterized by volumetric adsorption measurements and temperature-programmed desorption and reaction techniques. The reaction in both directions was observed to follow two parallel mechanistic pathways, namely formate- and oxygen vacancy-assisted mechanisms, with relative rate contributions depending on temperature and sample pretreatment.

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### 1. Introduction

It is well known that gallium oxides and Ga-containing materials are catalytically active in a wide variety of reactions, including hydrocarbon dehydrogenation and aromatization [1–3], hydrocarbon isomerization [4], reduction of NO<sub>x</sub> by hydrocarbons [5], methanol steam reforming [6,7], conversion of methanol to hydrocarbons [8], and carbon dioxide hydrogenation/methanol synthesis [9–11]. Different types of active Ga sites with respect to oxidation state and geometry have been postulated in the literature [2,12,13]. For Ga-doped H-ZSM5 zeolites, mobile Ga<sub>2</sub>O species interacting with the zeolite lattice have been discussed [12]. Ga-hydride species have been identified as active species during the aromatization of light alkanes [2]. In addition, for CO<sub>2</sub> hydrogenation, surface hydrides Ga<sup>δ+</sup>–H (δ < 2) are reported to play a dominant role in the formation and conversion of hydrogenated oxycarbonaceous reaction intermediates [13]. Furthermore, for the specific case of methanol steam reforming and synthesis, the exceptionally high activity of Ga<sub>2</sub>O<sub>3</sub>-supported Pd catalysts has been ascribed to the presence of Pd–Ga alloys forming during reductive treatments at temperatures around 523 K [6].

Because knowledge about the interaction of the reactants with the sample surface is crucial to understanding the reaction mechanism and identifying the nature of the intermediates and the catalytically active centers, numerous studies on the interaction of

important probe molecules with different Ga-oxide samples have been carried out [14–21]. In terms of methanol steam reforming and synthesis, these include H<sub>2</sub>, CO<sub>2</sub>, CO, H<sub>2</sub>O, and CH<sub>3</sub>OH. A series of contributions has focused on the interaction of H<sub>2</sub> and CO<sub>2</sub> with different Ga<sub>2</sub>O<sub>3</sub> polymorphs [10,11,13–16]. Regarding CO<sub>2</sub>, after a preceding oxidation/reduction cycle, CO<sub>2</sub> exposure at 323 K and subsequent heating in streaming CO<sub>2</sub> up to 723 K was found to lead to up to six differently adsorbed carbon-oxygenate species with different binding energies as identified by in situ FTIR spectroscopy [16]. Studies on the interaction of CO with  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> and other gallia polymorphs were aimed mainly at gaining insight into the surface acidity of solid gallia-containing catalysts and CO-sensing properties [17,18]. The interaction of H<sub>2</sub>O with  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> surfaces was examined in depth by Meixner et al. [19], with special focus on the co-adsorption of H<sub>2</sub>O with probe molecules relevant for the (inverse) water–gas shift reaction (WGSR), most notably H<sub>2</sub> and CO. A strong influence of the H<sub>2</sub>O adsorption by formation of surface OH groups on the co-adsorption of other molecules, on the structure of the material itself, and on the catalytically active centers was observed [19]. Finally, the adsorption of CH<sub>3</sub>OH molecules was studied in detail by Bonivardi et al. [20,21] through FTIR spectroscopy and DFT calculations. The local surface chemical composition was found to be most crucial, giving rise to three types of different adsorption pathways: nondissociative, dissociative, and oxidative decomposition [20,21].

Because the surface chemistry of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> has been identified as the crucial parameter affecting the adsorption properties of methanol synthesis/steam-reforming reactant molecules and in-

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intermediates, the present contribution provides a systematic study of the adsorption of CO<sub>2</sub> and CO on differently pretreated  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> samples with different surface chemistry and surface defect structures. Special attention is given to a precise quantification of the adsorbed molecules by volumetric adsorption measurements. Because the WGSR is of crucial importance in methanol synthesis/steam reforming, the influence of the surface chemistry and defect structure of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> samples on both routes of the WGSR (i.e., CO + H<sub>2</sub>O vs CO<sub>2</sub> + H<sub>2</sub>) is highlighted. This work builds on and extends our earlier paper dealing with H<sub>2</sub> and H<sub>2</sub>O adsorption and defect formation on  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> [22].

## 2. Experimental

### 2.1. Materials

Two different  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> samples were used in this work. First, a commercial low-surface area  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> sample (99.99% purity), supplied by Alfa Aesar, was calcined in air at 1000 K. The surface area after pretreatment of the samples was determined by adsorption of nitrogen at 77 K according to BET (area, 4.0 m<sup>2</sup>/g). In addition to the low-surface area  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> sample, a high-surface area sample was prepared by dissolution of the commercial sample using 30% aqueous NaOH in a PTFE beaker (to avoid alkaline disintegration of glass), followed by dilution with water and precipitation of crystalline Ga(OH)<sub>3</sub> at around 90 °C using an equimolar amount of ammonium nitrate (with respect to NaOH) in aqueous solution. The precipitate was thoroughly washed with hot deionized water, filtrated, and dried in air at 130 °C for 1 h. According to [18], the precipitation of a GaO(OH) containing gel from an aqueous solution followed by drying and calcination at 673 K leads to the exclusive formation of  $\alpha$ -Ga<sub>2</sub>O<sub>3</sub>, which can be converted to  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> by high-temperature annealing at 1073 K for 4 h. The latter heating procedure was thus applied in air to our sample, yielding a well-ordered  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> sample with a surface area of 19.4 m<sup>2</sup>/g. The comparable crystallographic order of both  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> samples was demonstrated by X-ray diffraction.

The “reprecipitated”  $\sim$ 20 m<sup>2</sup>/g sample also was evaluated by XPS to estimate the surface-related contaminants;  $\leq$ 3% Na was found in the near-surface regions at maximum. Thus, a limited influence of near-surface Na<sup>+</sup> species onto the measurements presented in this work was discernible. Differences between the both samples with respect to hydrogen adsorption were rather small. The total amount of adsorbed H<sub>2</sub> under at the same experimental conditions was almost the same, but the reducibility of the surface to form vacancies was somewhat lower for the Na-containing high-surface area sample (cf. [22]). Nevertheless, the general trends observed on H<sub>2</sub> adsorption were comparable. With respect to CO<sub>2</sub> adsorption, the expected difference of a higher number of adsorption sites for carbonate formation was verified, as discussed in Section 2.2.

Hydrogen (5.0), helium (5.0), and oxygen (3.5) were high-grade gases supplied by Messer-Griesheim. Hydrogen was further purified by passage through an oxygen-removing purifier. Condensable contaminants were removed from hydrogen and helium by liquid nitrogen traps, whereas oxygen was passed through a trap cooled with liquid nitrogen/ethanol. CO<sub>2</sub> was dried by passage through a molecular sieve and a liquid nitrogen/ethanol cooling trap at 223 K.

CO (at a maximum of 200 mbar) was dried over liquid nitrogen. Mixtures for the WGSRs were always 1:1 mixtures of CO and H<sub>2</sub>O with a total pressure of 48 mbar. For the inverse WGSRs, 1:1 mixtures of CO<sub>2</sub> and H<sub>2</sub> with a total pressure of 48 mbar also were used. These differently pretreated  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> samples were exposed to CO/H<sub>2</sub>O and CO<sub>2</sub>/H<sub>2</sub> mixtures, as outlined below.

**Table 1**

Defect concentration and H<sub>2</sub> coverage as a function of the reduction temperature and O<sub>2</sub> reuptake after H<sub>2</sub> desorption

H <sub>2</sub> -pretreatment temperature [K]	$\beta$ -Ga <sub>2</sub> O <sub>3</sub> 4.0 m <sup>2</sup> /g		$\beta$ -Ga <sub>2</sub> O <sub>3</sub> 19.4 m <sup>2</sup> /g	
	H <sub>2</sub> -adsorption [ $\mu$ mol/g]	O <sub>2</sub> -reuptake [ $\mu$ mol/g]	H <sub>2</sub> -adsorption [ $\mu$ mol/g]	O <sub>2</sub> -reuptake [ $\mu$ mol/g]
373	0.8	0.0	3.9	0.0
473	3.3	0.0	9.7	0.0
573	5.6	0.1	21.3	0.1
673	7.0	2.4	29.1	2.0
773	7.3	5.6	31.0	9.7
873	7.1	10.4	33.0	21.3

Reduction in a flow of 1 bar dry H<sub>2</sub> (liquid nitrogen trap), 60 ml/min, at the respective temperature for 1 h. The defect concentration was quantified in [22] by temperature-programmed volumetric oxidation.

### 2.2. Volumetric apparatus

Volumetric measurements were performed in an all-glass apparatus equipped with metal bellow valves (Witeg), a Baratron pressure transducer (MKS), and mass flow controllers (MKS). To ensure volumetric measurements at temperatures of up to 1000 K, the reactor part of the apparatus containing the sample was made of quartz glass. The mass of the gallia samples was 3.1611 g for the low-surface area sample and 0.9780 g for the high-surface area sample, to provide sufficient volumetric effects in the measurements.

Pretreatment of the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> surface was carried out in hydrogen at 298–773 K and in oxygen at 973 K at a flow rate of 1.0 mL/s. Before each oxygen treatment, the sample was evacuated at 300 K by a diffusion pump (base pressure,  $5 \times 10^{-7}$  mbar), heated in high-vacuum conditions to 973 K at a rate of 10 K/min (TPD measurements), and then cooled in vacuo. The oxygen uptake was measured volumetrically at 313–973 K to ensure quantitative reoxidation of gallia to its stoichiometric state.

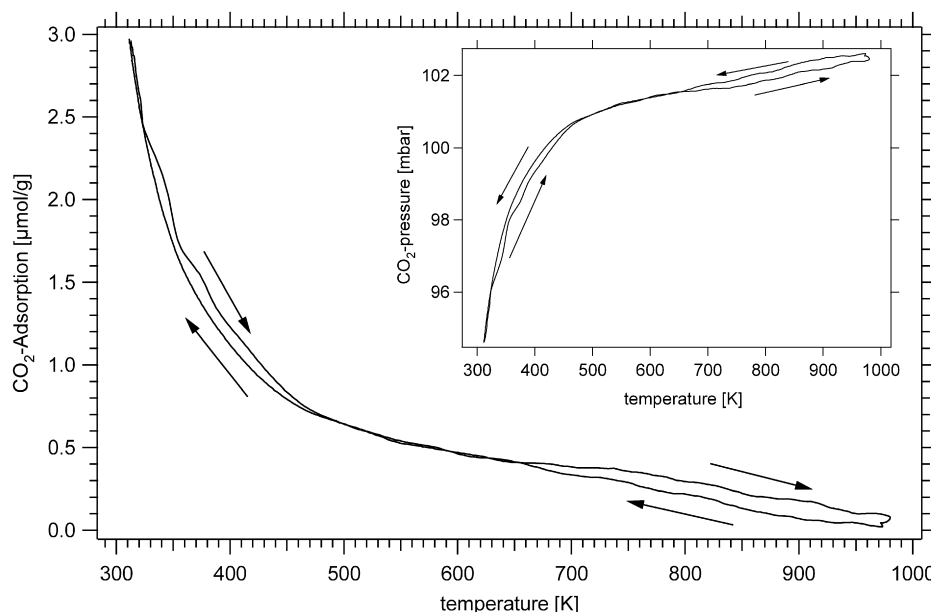
Reactant detection was carried out by continuous mass spectrometry detection. All of the temperature-programmed CO and CO<sub>2</sub> reactions, as well as the (inverse) WGSR, were corrected with respect to the thermal expansion of the reaction gas mixtures, and the  $m/z = 28$  CO mass spectrometry signal was corrected for the simultaneous contribution of the respective CO<sub>2</sub> fragment. Complete mass balance of both the WGSR and the inverse WGSR reaction routes was established by a very good agreement between the sum of the QMS-derived partial pressures of CO, CO<sub>2</sub>, H<sub>2</sub>, and H<sub>2</sub>O with the simultaneously recorded total pressure at any time and reaction temperature.

## 3. Results and discussion

### 3.1. H<sub>2</sub> and H<sub>2</sub>O adsorption and defect formation on $\beta$ -Ga<sub>2</sub>O<sub>3</sub>

As outlined in a previous article on H<sub>2</sub> adsorption and defect formation on  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> [22], it is experimentally possible to prepare defective surfaces of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> with and without adsorbed hydrogen (see Table 1). We particularly emphasize the importance of using dry H<sub>2</sub> (dried in a liquid nitrogen trap) for surface reduction and defect formation, to avoid defect quenching by heterolytic water adsorption and/or hydrolysis of H(ads) on reduced surface Ga centers. From the results in [22], it is clear that defects and H(ads) are highly sensitive to hydrolysis. We found that switching from dry H<sub>2</sub> to H<sub>2</sub> + 6.1 mbar H<sub>2</sub>O caused the immediate quenching of defects, as demonstrated by subsequent quantitative TPO analysis. The intensity of the Ga–H infrared region ( $\sim$ 2010 cm<sup>-1</sup>) was strongly diminished in the presence of small traces of water.

By using water-free H<sub>2</sub>, the defect concentration can be controlled via the reduction temperature and time, whereby a temperature of 550 K must be exceeded to initiate the reduction of



**Fig. 1.** Reversible  $\text{CO}_2$ -adsorption on fully oxidized  $\beta\text{-Ga}_2\text{O}_3$  ( $4\text{ m}^2/\text{g}$ ). Heating rate and cooling rate:  $10\text{ K}/\text{min}$ . The highest temperature ( $973\text{ K}$ ) was kept constant for  $10\text{ min}$ . Initial equilibrium  $\text{CO}_2$  pressure  $95\text{ mbar}$ , for better comparison with e.g. Fig. 4 or Fig. 6 the inset shows the temperature-corrected evolution of the  $\text{CO}_2$  pressure as a function of the reaction cell temperature.

$\text{Ga-O}$  bonds and the removal of lattice oxygen as water, which is constantly removed by using flowing dry  $\text{H}_2$ . Thus, the defective but  $\text{H}(\text{ads})$ -free  $\beta\text{-Ga}_2\text{O}_3$  sample can be prepared by removing the remaining  $\text{H}(\text{ads})$  via thermal desorption under high-vacuum conditions, and their  $\text{H}(\text{ads})$ -covered counterparts can be prepared by cooling in  $\text{H}_2$  without subsequent  $\text{H}_2$  thermal desorption.

In summary, four different chemical states of the  $\beta\text{-Ga}_2\text{O}_3$  surface can be deliberately prepared in a controlled manner to assess both the reactivity of the defects and of the adsorbed hydrogen toward  $\text{CO}$  and  $\text{CO}_2$  independently, which are potentially important processes in the WGSR:

- The fully oxidized (stoichiometric), water-depleted surface, using dry oxygen for oxidation.
- The hydrogen-covered surface without oxygen vacancies.
- The hydrogen-covered surface with a certain number of vacancies.
- The hydrogen-free surface with a certain number of vacancies.

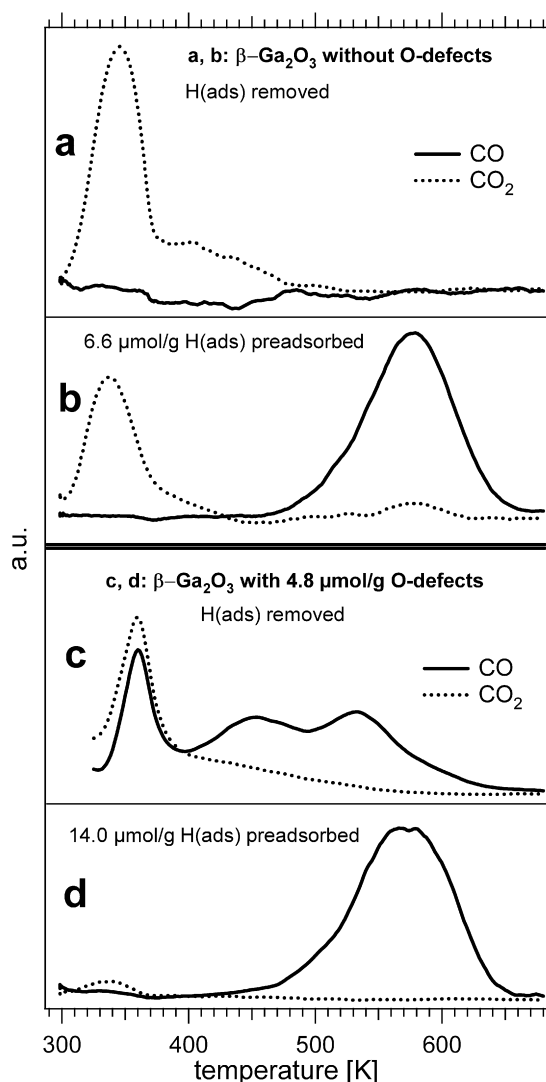
### 3.2. $\text{CO}_2$ adsorption

To characterize the differences in reactivity in the fully oxidized, hydrogen-free defective, and hydrogen-covered defective  $\beta\text{-Ga}_2\text{O}_3$  surfaces with respect to  $\text{CO}_2$ , first the fully oxidized surface was exposed to clean, dry  $\text{CO}_2$  in a TPR experiment. The volumetric adsorption of dry  $\text{CO}_2$  (equilibrium pressure of about  $100\text{ mbar}$ ) on fully oxidized  $\beta\text{-Ga}_2\text{O}_3$  at  $313\text{ K}$  resulted in an adsorbed amount of  $3.0\text{ }\mu\text{mol CO}_2/\text{g } \beta\text{-Ga}_2\text{O}_3$  ( $4.0\text{ m}^2/\text{g}$  sample; Fig. 1). Subsequently, the sample was subjected to a heating-cooling cycle that comprised heating from  $313$  to  $973\text{ K}$  (rate,  $10\text{ K}/\text{min}$ ), keeping the temperature constant at  $973\text{ K}$  for  $10\text{ min}$ , then cooling at a rate of  $-10\text{ K}/\text{min}$ . It was immediately seen that  $\text{CO}_2$  desorbed during heating, and that the same amount of  $\text{CO}_2$  re-adsorbed on  $\beta\text{-Ga}_2\text{O}_3$  during cooling; that is, a reversible desorption and adsorption of  $\text{CO}_2$  over the whole temperature range was observed. According to Collins et al. [16],  $\text{CO}_2$  can adsorb in various ways on  $\beta\text{-Ga}_2\text{O}_3$ . The relevant species for carbonate formation from  $\text{CO}_2$  include a weakly bound carboxylate species and monodentate, bidentate, and polydentate carbonates formed at  $313\text{ K}$  [16]. The different processes leading to the spectroscopically identified

species cannot be clearly discerned from the curves shown in Fig. 1; however, from Arrhenius plots of the low- and the high-temperature branches of the curves in Fig. 1, differently bound  $\text{CO}_2$  species—a weakly bound, physisorbed (heat of adsorption,  $10\text{ kJ}/\text{mol}$ ) and a more strongly bound  $\text{CO}_2$  species (heat of adsorption,  $60\text{--}80\text{ kJ}/\text{mol}$ )—can be distinguished on the  $4\text{-m}^2/\text{g}$  sample. It is clear that both  $\text{CO}_2$  physisorption and carbonate formation and decay are largely reversible over the whole temperature range, because no hysteresis effects are observed. Because the variation of the total pressure  $\text{CO}_2$  in the TPR recipient is only  $95\text{--}102.5\text{ mbar}$  (T-uncorrected,  $112\text{ mbar}$ ), this means that the data of Fig. 1 basically can be interpreted as a close approximation of an adsorption isobar.

Additional Na-induced basic surface centers were likely responsible for the increased number of  $\text{CO}_2$  adsorption sites forming carbonates in the  $19.4\text{-m}^2/\text{g}$  sample. The maximum adsorbed amount of  $\text{CO}_2$  on the high-surface area sample at  $313\text{ K}$  in the same pressure range was  $2.4\text{ }\mu\text{mol}/\text{m}^2$ , corresponding to  $46\text{ }\mu\text{mol}/\text{g}$ , compared with  $0.75\text{ }\mu\text{mol}/\text{m}^2$ , corresponding to  $3\text{ }\mu\text{mol}/\text{g}$ , on the low-surface area sample. The isobaric adsorption of  $\text{CO}_2$  was measured on both samples and was found to be very similar with respect to the temperature-dependent adsorption characteristics, indicating that the major difference is in the total surface density of available sites. Whether this difference is caused mainly by Na or by surface structural differences remains an open question. The heats of adsorption on the high-surface area gallia sample are nearly the same as those obtained on the  $4\text{-m}^2/\text{g}$  sample (weakly bonded  $\text{CO}_2$ ,  $10\text{ kJ}/\text{mol}$ ; strongly bonded  $\text{CO}_2$ ,  $\sim 65\text{ kJ}/\text{mol}$ ), indicating that the Na impurities have no significant influence on the qualitative binding behavior of  $\text{CO}_2$ .

To further clarify the WGSR mechanism, TPD of  $\text{CO}_2$  experiments were carried out on four differently prepared  $\beta\text{-Ga}_2\text{O}_3$  surfaces: surfaces prerduced at  $473$  and  $673\text{ K}$  (i.e., no oxygen vacancies vs vacancy formation), each modified or not modified by adsorbed hydrogen (Fig. 2). Generally, after each pretreatment,  $\text{CO}_2$  was exposed for  $30\text{ min}$  at  $313\text{ K}$  resulting in an equilibrium pressure of approximately  $100\text{ mbar CO}_2$ . Subsequently, high vacuum was applied to the samples to remove the weakly bound (physisorbed)  $\text{CO}_2$ . Thus, subsequent TPD only revealed the more strongly adsorbed chemisorbed species that remained adsorbed in



**Fig. 2.** TPD spectra of CO and CO<sub>2</sub> obtained after CO<sub>2</sub>-adsorption on prereduced  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (4 m<sup>2</sup>/g): (a) and (b) prereduced at 473 K (i.e. without oxygen defects), (c) and (d) prereduced at 673 K (i.e. 4.8  $\mu$ mol/g oxygen vacancies present). (a) Without oxygen defects and without H(ads), (b) without oxygen defects but with 6.6  $\mu$ mol/g H(ads) preadsorbed. Full lines: CO traces, dashed lines: CO<sub>2</sub> traces. (c) With oxygen defects but without H(ads), (d) with oxygen defects and with 14.0  $\mu$ mol/g H(ads) preadsorbed. Full lines: CO traces, dashed lines: CO<sub>2</sub> traces.

vacuo at 313 K. The TPD experiment included a linear temperature ramp from 313 to 973 K at a rate of 10 K/min. Figs. 2a and 2b show the respective CO<sub>2</sub> and CO-TPD curves without oxygen vacancies present, that is, after prereduction at 473 K for 1 h, followed either by cooling in H<sub>2</sub> to 313 K and high vacuum for 20 min at 313 K [H(ads) present; Fig. 2b] or by heating in high vacuum up to 973 K at a rate of 10 K/min, followed by cooling in high vacuum to 313 K [H(ads) removed; Fig. 2a]. The respective defect and hydrogen surface concentrations can be deduced from Table 1. According to [22] and Table 1, the desorption of hydrogen from the vacancy-free sample prepared at 473 K eventually resulted in the reestablishment of the initial fully oxidized surface state without oxygen vacancies.

Exposure of dry CO<sub>2</sub> after removal of H(ads) at 313 K resulted in CO<sub>2</sub> adsorption of 2.9  $\mu$ mol/g. According to the subsequent TPD spectra (indicated by the dashed line in Fig. 2a), CO<sub>2</sub> was the only desorption product, and only formation/thermal decomposition of carbonate species (i.e., the same behavior as seen on the original, fully oxidized  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> system) was observed.

An oxygen vacancy-free surface with 6.6  $\mu$ mol H(ads)/g was prepared by a H<sub>2</sub> pretreatment at 473 K for 1 h, followed by cooling in H<sub>2</sub> to 313 K and high-vacuum exposure for 20 min before admission of dry CO<sub>2</sub> (Fig. 2b). According to [14] (see Fig. 5) and [22], after H<sub>2</sub> reduction at 473 K, both the presence of homolytically adsorbed H<sub>2</sub> on terminal oxygen species as –OH and the partial formation of Ga–H species via heterolytic H<sub>2</sub> adsorption [14] (with both types adsorbed on the vacancy-free surface) was observed. Adsorption of dry CO<sub>2</sub> at 313 K onto this surface resulted in 1.1  $\mu$ mol/g of adsorbed CO<sub>2</sub>. In the respective TPD (indicated by the solid line in Fig. 2b), a CO-peak with a maximum at about 580 K was observed, indicating that part of the adsorbed CO<sub>2</sub> was able to react with H(ads) in the absence of oxygen defects directly after adsorption or during the subsequent temperature ramp of the TPD experiment. As we explain later in the context of Fig. 6, the interaction of reactive Ga–H groups on this surface most likely led to the formation of formate species, which could subsequently react toward CO and adsorbed –OH groups. The formation of bidentate HCOO species from H<sub>2</sub> and CO<sub>2</sub> at temperatures above 450 K and their decay at temperatures above 550 K have been reported by Collins et al. [11]. This mechanism, termed the “formate mechanism” in what follows, may be active in either direction, depending on the initially established surface and reaction conditions. A CO<sub>2</sub> peak observed in the low-temperature region around 350 K indicates that the part of the CO<sub>2</sub> that was not converted to CO was adsorbed as a rather weakly bound carbonate species.

Figs. 2c and 2d show the TPD spectra from the surface preparations with 4.8  $\mu$ mol/g of oxygen vacancies present. Without H(ads) (after desorption of H<sub>2</sub> under high-vacuum conditions at 673 K and subsequent heating in high vacuum up to 973 K with +10 K/min, followed by cooling down in high-vacuum to 313 K), exposure of dry CO<sub>2</sub> at 313 K led to adsorption of 3.2  $\mu$ mol/g of CO<sub>2</sub> at 313 K (Fig. 2c). In the subsequent TPD spectra, desorption of both CO<sub>2</sub> and CO in the low-temperature region at around 350 K was observed. First, a CO<sub>2</sub> peak in the low-temperature region around 350 K indicates that the part of the CO<sub>2</sub> that was not converted to CO was again adsorbed as a rather weakly bound carbonate species. Second, the observation of a series of CO desorption maxima (starting at  $\sim$ 350 K) means that  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> modified with O-vacancies was able to react with CO<sub>2</sub> toward CO at slightly above room temperature, leading to partial quenching of oxygen defects. Apparently, three different CO-binding states were generated through this quenching process, which obviously does not require adsorbed hydrogen-containing species, such as formate.

The last case that we evaluated involves the adsorption and reaction of CO<sub>2</sub> on a  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> with both oxygen vacancies and H(ads) present, prepared as outlined earlier (Fig. 2d). Exposing this surface to dry CO<sub>2</sub> at 313 K resulted in the adsorption of 1.1  $\mu$ mol/g of CO<sub>2</sub>. A large CO peak with a maximum at about 570 K was observed, very similar to the CO peak resulting from the H(ads)-covered surface without defects (Fig. 2b, solid line). Only very little desorption of CO<sub>2</sub> stemmed from some residual carbonates, in contrast to the situation for the H(ads)-covered surface without defects, on which a considerably larger CO<sub>2</sub> peak was observed in the TPD spectra. This finding indicates a more complete conversion of CO<sub>2</sub> toward CO compared with the initial state without vacancies but otherwise a similar situation involving surface formates, with most of the desorbed CO apparently resulting from formate decomposition.

In summary, compared with reversible CO<sub>2</sub> adsorption as carbonates on a fully oxidized surface, the situation changed drastically if the initial state of the surface was oxygen-defective and/or hydrogen-covered. In accordance with [11], H(ads) is expected to react with CO<sub>2</sub> toward formate species (essentially bridged bidentate HCOO), which decay toward CO and surface OH groups at temperatures above 550 K, as evidenced by the predominant CO formation shown in Fig. 2. Even more active for reductive CO<sub>2</sub> ac-



tivation are the “hydrogen-free” oxygen vacancies, which enable CO formation in the low-temperature region ( $T > 300$  K). In all cases shown in Fig. 2, at least a small part of the CO<sub>2</sub> was adsorbed in the carbonate form, leading to a CO<sub>2</sub>-TPD feature with a rate maximum around 350 K.

The smallest CO<sub>2</sub> contribution, as well as the complete absence of low-temperature (<400 K) CO desorption features, was observed on the sample with H(ads)-blocked vacancies (which may alternatively be denoted as *cus* Ga–H sites); see Fig. 2d. It is likely that this sample did not allow for a direct reaction of residual H(ads)-free vacancies with CO<sub>2</sub>, because these sites were largely blocked by H(ads). Only the hydrogen-depleted sample, exhibiting an equivalent amount of *cus* Ga sites without H(ads), could effectively create the weakly bonded CO shown in Fig. 2c. From Fig. 2, it appears that all kinds of Ga–H species (i.e., *cus* Ga–H from homolytically adsorbed hydrogen at *cus* Ga sites and/or also Ga–H resulting from heterolytic H<sub>2</sub> adsorption without vacancies, see [22]) tended to produce formate in the presence of CO<sub>2</sub>.

From previous measurements [22], we cannot determine whether the oxygen vacancies created during reductive pretreatment were actually quantitatively occupied by H(ads) or whether some of them remained free *cus* Ga sites. But comparing Figs. 2c and 2d seems to indicate that the *cus* Ga sites were largely H(ads)-blocked before CO<sub>2</sub> exposure in Fig. 2d.

Two possible explanations exist for the very small amount of carbonate-related CO<sub>2</sub> in remaining in the sample shown in Fig. 2d: (a) more efficient site blocking for carbonate formation, caused by the much higher total H(ads) coverage on this sample (total hydrogen surface concentration, 14 μmol/g), and (b) more efficient conversion of intermediate carbonates toward formate species due to the higher amount of coadsorbed hydrogen at or slightly above room temperature. Support for either possibility comes from the CO<sub>2</sub> adsorption on the low-surface area gallia sample at around room temperature; at 313 K, on the fully oxidized, H(ads)-free surface, 3.0 μmol/g CO<sub>2</sub> adsorbed, whereas on the 14 μmol/g H(ads)-precovered surface, only ~1.1 μmol/g CO<sub>2</sub> adsorbed. Therefore, a combination of partial blocking of reversible carbonic CO<sub>2</sub> adsorption sites H(ads) and efficient conversion of the intermediately adsorbed CO<sub>2</sub> (e.g., as weakly bonded carbonate species) toward formate species (equally blocking sites) appears likely. The finding that Fig. 2d shows almost quantitatively CO and that Fig. 2b also shows some remaining CO<sub>2</sub> still bonded as carbonate thus likely can be explained by the higher total hydrogen surface concentration on the initially present vacancy + H(ads) sample (14 μmol/g in Fig. 2d vs 6.6 μmol/g in Fig. 2b), that is, more efficient conversion toward formate and/or site blocking. Based on our own experiments, we cannot distinguish between the surface reaction of adsorbed hydrogen with particular intermediate carbonates and/or the (unlikely) direct reaction of CO<sub>2</sub>(g) with the Ga–H reactive sites. Rather, according to [11], it seems that the major part of the carbonate species are first decomposed and/or converted toward formate at temperatures up to 450 K. Our TPD spectra given in Fig. 2 also show that most of the carbonates were desorbed up to this temperature. A simplified picture can be suggested in which the intermediate surface concentration of weakly bonded carbonates is high at low temperatures (<350 K) and the conversion toward formate species remains slow at these temperatures. A temperature rise to above 400 K increases the decomposition of the intermediate carbonate species as well as the rate of conversion to formate; that is, their presence in the spectra is strongly diminished. However, even at a low steady-state coverage, these species may represent the dominant kinetic intermediate from CO<sub>2</sub> and H(ads) toward formate. The reaction scheme shown in Fig. 5 does not differentiate these different mechanisms and thus does not include the potential carbonate intermediate.

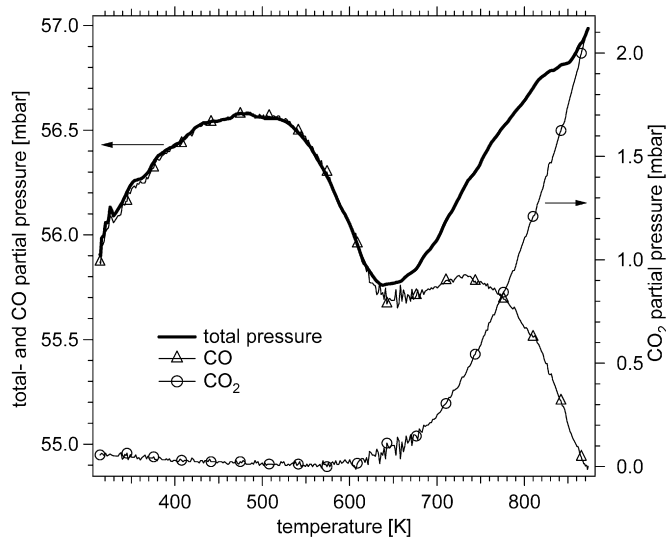


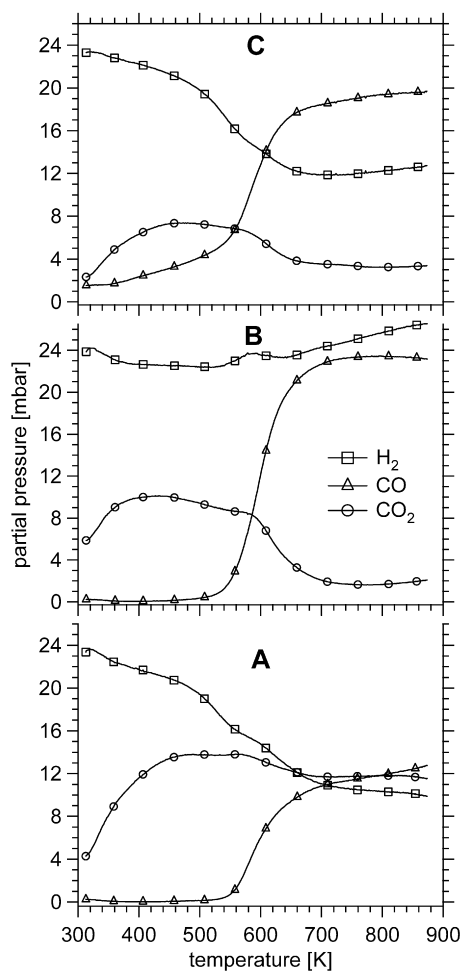
Fig. 3. CO-TPR on water-free  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (19.4 m<sup>2</sup>/g). Initial CO pressure ~56 mbar. Heating rate: 10 K/min.

### 3.3. CO adsorption and reaction

Because dry CO<sub>2</sub> is capable of quenching oxygen vacancies in the absence of water or surface OH species, we anticipated that the reverse process—defect formation using dry CO—also should be feasible. The TPR of initially fully oxidized  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> with dry CO is shown in Fig. 3. The preceding volumetric CO adsorption was carried out at 313 K on the fully oxidized  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (19.4 m<sup>2</sup>/g), resulting in an equilibrium pressure of about 56 mbar CO and 1.5 μmol/g of adsorbed CO. Then the sample was heated to 973 K (heating rate, 10 K/min). During heating between 300 and 500 K, weakly (obviously molecularly) bonded CO desorbed. Starting at ~500 K, CO adsorbed again due to a surface reaction consuming CO. This reaction most likely created reactively adsorbed CO products at reduced Ga centers. Because at above 500 K, adsorption of CO occurred without simultaneous CO<sub>2</sub> formation in the gas phase, we conclude that the CO remained in this strongly chemisorbed state, possibly carbonate species close to oxygen vacancies, up to 630 K, at which point CO<sub>2</sub> desorption started by decomposition of this state. Once CO<sub>2</sub> desorbed to the gas phase, oxygen vacancies necessarily remained on the surface. This experiment proves that a second reaction mechanism for the WGS is feasible above 630 K; we call this the “vacancy-mediated” mechanism. This mechanism involves formation of defects through simultaneous CO and CO<sub>2</sub> formation and the continuous reoxidation of these defects by water, inducing H<sub>2</sub> production through a full catalytic cycle.

### 3.4. (Inverse) WGS

Finally, the temperature-dependent progress of both the inverse WGS and the WGS were investigated through TPR experiments in either direction. To assess the influence of the initial surface state, identical TPR runs were started from the fully oxidized, hydrogen-covered, and vacancy-modified surfaces without H(ads) (Figs. 4A–4C). The reactivity-versus-temperature plot starting from the H(ads)-covered but vacancy-free surface turned out to be very similar to that highlighted in Fig. 4B [H(ads) + vacancies] and thus is not shown. This is likely because only the initial presence of vacancies without H(ads) allowed for direct interaction of some of the CO<sub>2</sub> with these (already at around 300 K) to form weakly bonded CO desorbing at low temperatures (i.e., <400 K; see Fig. 2c); if H(ads) were initially covering the vacancies, then



**Fig. 4.** Inverse WGS activity as a function of temperature on  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (0.978 g of the 19.4 m<sup>2</sup>/g sample). (A) Adsorption of a 1:1 CO<sub>2</sub>/H<sub>2</sub>-mixture on fully oxidized  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> at 313 K. The total amount of introduced gas was 48 mbar at 313 K, whereby 20.5 mbar gas was adsorbed already at 313 K, mostly CO<sub>2</sub>. (B) Same mixture adsorbed at 313 K onto the 773 K hydrogen-covered defective sample (64  $\mu$ mol/g H(ads) + 20.6  $\mu$ mol/g O-vacancies) at 313 K. Total amount of introduced reactant gas 48 mbar, 18 mbar gas adsorbed at 313 K, mostly CO<sub>2</sub>. (C) Same mixture adsorbed at 313 K onto 773 K prerduced and hydrogen-desorbed  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> sample, i.e. with an initial defect concentration of 20.6  $\mu$ mol/g O-vacancies. Total amount of introduced gas mixture again 48 mbar at 313 K, 21 mbar gas adsorbed, mainly CO<sub>2</sub>. The water signal was not included due to detection- and quantification problems. Linear heating ramp: 10 K/min.

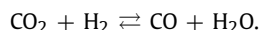
this direct CO<sub>2</sub> activation at defects toward CO obviously would be blocked (see Fig. 2d).

In Fig. 4A, the temperature-programmed inverse WGS was performed directly after adsorption of a 1:1 CO<sub>2</sub>:H<sub>2</sub> reaction mixture onto the fully oxidized  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> at 313 K, by heating to 873 K at a constant rate of 10 K/min. The major part of the introduced CO<sub>2</sub> was initially adsorbed at 313 K (24 mbar dosed, ~4 mbar remaining in the gas phase), and hardly any H<sub>2</sub> was adsorbed (23.5 mbar remaining). This finding can be explained by results of our group demonstrating that H<sub>2</sub> adsorption below 730 K is an activated process [22], in contrast to CO<sub>2</sub> adsorption (see Fig. 1). During heating from 313 K to slightly below 450 K, no formation of CO was observed, but a combination of activated hydrogen adsorption (decreasing H<sub>2</sub> pressure) and desorption of weakly bound CO<sub>2</sub> (increasing CO<sub>2</sub> pressure between 313 and 450 K) was seen.

The more weakly bound fraction of the carbonate species, according to [16], is likely to desorb up to a maximum of 500 K, as also can be deduced from Figs. 1 and 4. In fact, an increase in CO<sub>2</sub> pressure and a decrease in hydrogen pressure were found

to occur only up to ~450 K. According to [11], the conversion of adsorbed CO<sub>2</sub> and hydrogen toward a bridge-bonded formate species (designated bidentate HCOO in [11]) occurred at around this temperature. Accordingly, the decreasing trend of the H<sub>2</sub> pressure grew stronger, due to simultaneous consumption by adsorption and bidentate HCOO formation. The CO<sub>2</sub> pressure remained almost constant above 450 K, due to rate-balancing between consumption toward formate and desorption.

At around 550 K, the rate of CO formation increased steeply. Obviously, at this temperature, an additional process set in, leading to both increased CO<sub>2</sub> consumption and pronounced CO formation. It appears that at above 550 K, an additional decrease of the hydrogen pressure occurred by consumption through the reaction



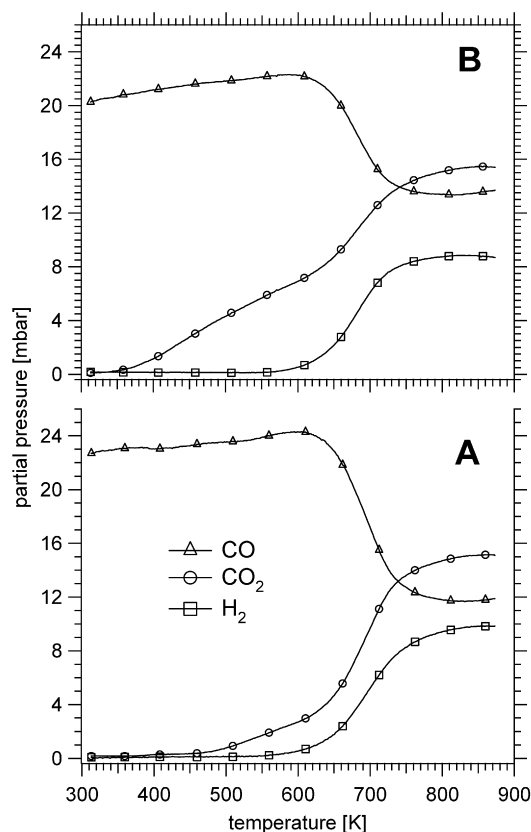
At 550 K, two effects were noted that have been previously described in the literature [11,22]: the onset of oxygen defect formation through hydrogen surface reduction and the formation of major amounts of bidentate HCOO species on the surface, resulting from simultaneous exposure to CO<sub>2</sub> and H<sub>2</sub> (see Fig. 5 in Ref. [11]). Of course, the onset of oxygen vacancy formation implies the possibility that the CO<sub>2</sub> was readily activated at these vacancies and reacted toward CO by quenching them, a process that starts temperatures below 400 K (cf. Fig. 2c, the reaction of CO<sub>2</sub> with hydrogen-free defects). Based on the information available so far, two distinct reaction mechanisms for the inverse WGS (and, conversely, the WGS; see Fig. 6) can be formulated: the formate mechanism (FM) and the vacancy-mediated mechanism (VMM). The latter is apparently a Mars-van Krevelen-type mechanism [23]. These two alternative mechanisms are depicted in Fig. 5 as operating in parallel, with temperature-variable contributions to the overall conversion.

Which mechanism is then more likely to contribute to the rate increase observed in Fig. 4A at temperatures above 550 K? According to Fig. 5, the first step of the inverse WGS is the same for both mechanisms—homolytic hydrogen adsorption followed by heterolytic hydrogen adsorption at the surface. From the experiments shown in [22] and in Table 1 of this work, we know that hydrogen is adsorbed without defect formation at temperatures below 550 K and that measurable amounts of oxygen vacancies are formed above this temperature. These are highly active in CO<sub>2</sub> activation and formation of CO via defect quenching, a process that occurs with dry CO<sub>2</sub> (cf. Fig. 2). The CO TPD spectra in Fig. 2 show that the main part of this “defect-generated” CO in fact desorbs at 550 K and below; therefore, we may conclude that, in principle, CO formed through the VMM can contribute to the increased inverse WGS reaction rate observed at temperatures above 550 K.

On the other hand, a certain contribution of the FM cannot be excluded, according to the IR spectra obtained on  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> in a H<sub>2</sub>-CO<sub>2</sub> reaction mixture by Collins et al. The relevant set of spectra in [11] shows that the bidentate HCOO intensity (with this species verified by a control experiment on our sample, to connect to these measurements) increases up to ~550 K but decreases thereafter, indicating thermal decomposition of the bidentate HCOO species to CO and OH(ads), as reported in [11]. The fact that the nondefective H(ads)-covered surface yielded a large amount of CO from CO<sub>2</sub> at around 570 K (see Fig. 2) suggests extensive decomposition of the formates toward CO.

Fig. 4B shows the course of the inverse WGS reaction starting from the hydrogen-covered defective surface. In analogy to the situation depicted for case A, initially little H<sub>2</sub> but most of the CO<sub>2</sub> was adsorbed, for the reasons discussed previously. The H<sub>2</sub> partial pressure remained high throughout the entire experiment, due to the absence of H<sub>2</sub> consumption by activated adsorption or, conversely, initial surface saturation with H(ads). Above 550 K, the inverse WGS set in, exactly at the same temperature as in case A.





**Fig. 6.** WGS activity as a function of temperature on  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (0.978 g of the 19.4 m<sup>2</sup>/g sample). (A) Adsorption of a 1:1 CO-H<sub>2</sub>O-mixture (sequentially, together 48 mbar gas in the reaction cell at 313 K) on fully oxidized  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> at 313 K. 25 mbar gas become immediately adsorbed at 313 K. All H<sub>2</sub>O is adsorbed, but only a small fraction of CO. (B) Adsorption of a 1:1 CO/H<sub>2</sub>O-mixture on defective  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (19.4 m<sup>2</sup>/g, pre-reduced at 773 K and H(ads) thermally desorbed leading to  $\sim$ 21  $\mu$ mol/g hydrogen-free O-vacancies) at 313 K. Total amount of adsorbed CO/H<sub>2</sub>O-mixture again 48 mbar, whereby 28 mbar gas become initially adsorbed. All H<sub>2</sub>O is adsorbed, but only a small fraction of CO. The H<sub>2</sub>O signal was not included due to detection- and quantification problems. Linear heating ramp: 10 K/min.

results presented in Fig. 2 after adsorption of CO<sub>2</sub> onto a hydrogen-reduced catalyst.

If the hydrogen was desorbed after reduction in the usual way (“hydrogen-free defects”), then CO<sub>2</sub> adsorbed at room temperature desorbed at temperatures up to 450 K as both CO and CO<sub>2</sub> at an approximate 1:1 ratio (Fig. 2c), whereas the hydrogen-precovered catalyst demonstrated hardly any CO desorption and some CO<sub>2</sub> desorption in the same temperature range (Fig. 2d). In general, the data in Fig. 2 show a considerably higher fraction of CO<sub>2</sub> desorbing below 450 K and a much larger fraction of CO desorbing above 450 K, even in the complete absence of water.

Of course, the major difference between the experiments shown in Figs. 2 and 6 is the simultaneous presence of CO and H<sub>2</sub>O on the catalyst surfaces in the experiment shown in Fig. 6. As discussed in the previous section and in [22], H<sub>2</sub>O is capable of partially re-oxidizing reduced  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> already at 313 K [22], whereby oxygen vacancy quenching creates a small fraction of gaseous H<sub>2</sub> and most likely a major fraction of heterolytically adsorbed water (i.e., -OH groups) at the initially present vacancy sites. Therefore, it can be safely assumed that the propensity of this water-quenched surface for CO<sub>2</sub>- rather than CO-desorption at temperatures below 450 K would be even higher than that of a hydrogen-terminated but water-free surface. Provided that the initial presence of defects promoted heterolytic water adsorption, a larger amount of -OH groups also could be formed on the surface at close to room temperature. We propose that these groups are particularly active

for CO binding and conversion toward CO<sub>2</sub> at above 300 K via the FM (i.e., the WGS direction; see the left side of Fig. 5).

This mechanism, which is apparently noncatalytic in Fig. 6 (because no H<sub>2</sub> is formed up to 570 K), will continuously increase the hydrogen surface coverage and the CO<sub>2</sub> gas-phase pressure until the desorption temperature of H<sub>2</sub> is reached. Comparing the hydrogen evolution curves of Figs. 6A and 6B shows that H<sub>2</sub> desorption began at 570–580 K in both cases, irrespective of the initial surface state, meaning that above this temperature, in principle, the FM could become catalytic. The marked additional increase in the CO<sub>2</sub> partial pressure slope observed above 630 K indicates an additional rate enhancement, which we assign to additional defect formation by the direct reaction of CO with lattice oxygen, according to the interpretation of Fig. 3, where at above  $\sim$ 630 K, CO began to reduce the  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> in a direct reaction channel (also in the absence of H<sub>2</sub>O/OH). Necessarily, consumption of CO from the gas phase also started above 630 K. Therefore, it appears that above 630 K, two different reaction channels for CO<sub>2</sub> formation were active simultaneously, with the low-temperature FM being overtaken at above 630 K by the probably more strongly activated VMM.

Applying the same concept to Fig. 6A (i.e., the fully oxidized initial state), the only difference is the smaller contribution of the low-temperature FM mechanism, starting in A at above 450 K. We can tentatively attribute this difference to the less efficient heterolytic water activation on the initially fully oxidized surface, leading to formation of fewer reactive OH groups at higher temperatures. The onset of H<sub>2</sub> desorption and CO-induced defect formation were observed at the same temperatures in the experiments shown in Figs. 6A and 6B.

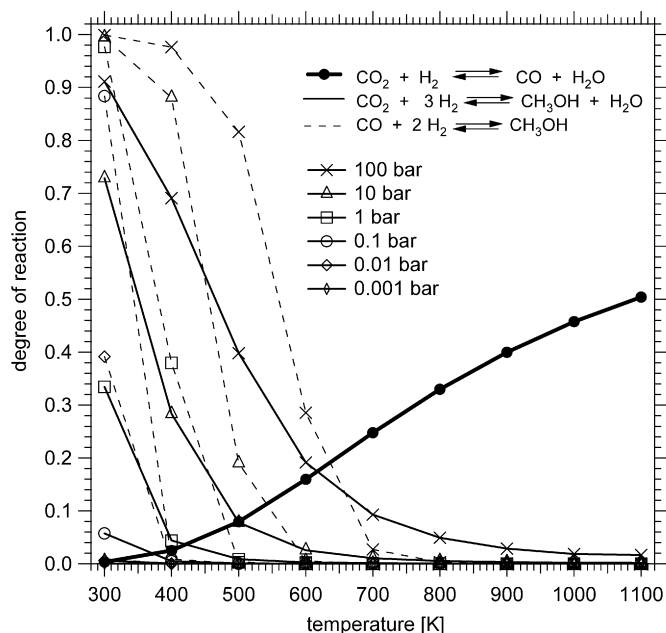
After all WGS runs, all of the initially present oxygen vacancies were quenched, and no measurable O<sub>2</sub> reuptake was observed. Because the vacancy quenching process was active already at  $\sim$ 300 K, we can conclude that most of the quenching occurred between 300 and 600 K, leading to the aforementioned hydroxyl species, and, therefore, the initial defect concentration had no effect on the mechanism at temperatures above 630 K.

Nevertheless, we consider the WGS reaction at above 630 K a truly cyclic catalytic process, because oxygen vacancies are continuously formed with CO to yield CO<sub>2</sub> and are quenched by water to yield hydrogen, leading eventually to a rather low stationary concentration of oxygen vacancies. The initially preformed defects influence only the “noncatalytic” part of the surface reaction at the lower temperatures (e.g., if they decompose a part of the water to form OH groups, which then can react with CO). In this case, the conversion of reactants/formation of products is no longer stoichiometric, and we consider this a noncatalytic surface reaction closely related to the “true” catalytic WGS.

#### 4. Conclusion

Our conclusions refer directly to the optimum attainable selectivity of the methanol steam-reforming reaction toward CO<sub>2</sub> and H<sub>2</sub> as a function of temperature on Ga<sub>2</sub>O<sub>3</sub>-supported catalysts. The mechanistic basis for considerations related to the selective reformation on Ga<sub>2</sub>O<sub>3</sub> and Pd/Ga<sub>2</sub>O<sub>3</sub> has been provided by the work of Collins et al. on the intermediates of methanol synthesis from carbon dioxide and hydrogen [11], which also describes the reverse process, according to the principle of microscopic reversibility. We may first address the question why the temperature-dependent balance of the WGS and methanol steam-reforming reaction is so important to catalytic selectivity. According to our work, temperatures of at least 600 K are needed to catalyze the equilibration of the (inverse) WGS at a high rate on pure Ga<sub>2</sub>O<sub>3</sub>, and thus, in principle, a high selectivity for the CO<sub>2</sub>-methanol pathway can





**Fig. 7.** Pressure-dependent reaction equilibria for three important reactions  $\text{CO}_2 + \text{H}_2 \rightleftharpoons \text{CO} + \text{H}_2\text{O}$ ,  $\text{CO}_2 + 3\text{H}_2 \rightleftharpoons \text{CH}_3\text{OH} + \text{H}_2\text{O}$  and  $\text{CO} + 2\text{H}_2 \rightleftharpoons \text{CH}_3\text{OH}$ , calculated from the respective thermodynamic data in [24].

be attained at lower temperatures without major CO formation via the WGSR equilibrium.

Fig. 7 displays the calculated equilibrium degree of reaction of WGSR, methanol synthesis/steam reforming, and CO hydrogenation/methanol dehydrogenation as a function of temperature and total pressure, as calculated from the respective thermodynamic data derived from [24] (not considering fugacities and condensation effects). From this figure, it is obvious that from a thermodynamic standpoint, in principle the reaction degree of the methanol steam reforming process can reach almost 100% in the low-pressure range (maximum 100 mbar) and at temperatures above 400 K. Iwasa et al. [6] examined methanol steam reforming toward  $\text{CO}_2$  and  $\text{H}_2$  over  $\text{PdGa}_x/\text{Ga}_2\text{O}_3$  at pressures around

100 mbar and a maximum temperature of  $\sim 500$  K (well below 600 K) and found a very high selectivity toward  $\text{CO}_2$  ( $\sim 95\%$ , i.e., only 5% CO at maximum) [6]. According to the results of our present study, we suggest that efficient catalytic equilibration of the WGSR on the clean  $\beta\text{-Ga}_2\text{O}_3$  support occurred above 600 K, whereby CO formation likely proceeded through the FM and VMM mechanisms, with relative contributions depending on the reaction temperature, and could induce up to 20% CO at around 600 K (see Fig. 7).

## References

- [1] I. Takahara, M. Saito, M. Inaba, K. Murata, *Catal. Lett.* 96 (2004) 29.
- [2] G.D. Meitzner, E. Iglesia, J.E. Baumgartner, E.S. Huang, *J. Catal.* 140 (1993) 209.
- [3] I. Nowak, J. Quartararo, E.G. Derouane, J.C. Vedrine, *Appl. Catal. A* 251 (1) (2003) 107.
- [4] J.A. Moreno, G. Poncelet, *J. Catal.* 203 (2001) 453.
- [5] K. Shimizu, A. Satsuma, T. Hattori, *Appl. Catal. B* 16 (1998) 319.
- [6] N. Iwasa, T. Mayanagi, N. Ogawa, K. Sakata, N. Takezawa, *Catal. Lett.* 54 (1998) 119.
- [7] N. Iwasa, T. Mayanagi, W. Nomura, M. Arai, N. Takezawa, *Appl. Catal. A* 248 (2003) 153.
- [8] D. Freeman, R.P.K. Wells, G.J. Hutchings, *J. Catal.* 205 (2002) 358.
- [9] T. Fujitani, M. Saito, Y. Kanai, T. Watanabe, J. Nakamura, T. Uchijima, *Appl. Catal. A* 125 (1995) L199.
- [10] A.L. Bonivardi, D.L. Chiavassa, C. Querini, M.A. Baltanas, *Stud. Surf. Sci. Catal.* 130D (2000) 3747.
- [11] S.E. Collins, M.A. Baltanas, A.L. Bonivardi, *J. Catal.* 226 (2004) 410.
- [12] P. Meriaudeau, C. Naccache, *Appl. Catal.* 73 (1991) L13.
- [13] S.E. Collins, M.L. Baltanas, J.L. Garcia Fierro, A.L. Bonivardi, *J. Catal.* 211 (2002) 252.
- [14] S.E. Collins, M.A. Baltanas, A.L. Bonivardi, *Langmuir* 21 (2005) 962.
- [15] E.A. Gonzalez, P.V. Jasen, A. Juan, S.E. Collins, M.A. Baltanas, A.L. Bonivardi, *Surf. Sci.* 575 (2005) 171.
- [16] S.E. Collins, M.A. Baltanas, A.L. Bonivardi, *J. Phys. Chem.* 110 (2006) 5498.
- [17] T. Schwebel, M. Fleischer, H. Meixner, C.-D. Kohl, *Sens. Actuators B* 49 (1998) 46.
- [18] M. Rodríguez Delgado, C. Otero Arean, *Z. Anorg. Allg. Chem.* 631 (2005) 2115.
- [19] F. Reti, M. Fleischer, H. Meixner, *J. Giber, Sens. Actuators B* 18–19 (1994) 138.
- [20] M.M. Branda, S.E. Collins, N.J. Castellani, M.A. Baltanas, A.L. Bonivardi, *J. Phys. Chem. B* 110 (2006) 11847.
- [21] S.E. Collins, M.A. Baltanas, A.L. Bonivardi, *Appl. Catal. A* 295 (2005) 126.
- [22] W. Jochum, S. Penner, K. Föttinger, R. Kramer, G. Ruppel, B. Klötzer, *J. Catal.* (2008), doi: 10.1016/j.jcat.2008.03.019, in press.
- [23] P. Mars, D.W. van Krevelen, *Chem. Eng. Sci.* 3 (1954) 41.
- [24] I. Barin, *Thermodynamic Data of Pure Substances*, VCH, Weinheim, 1993.